# High Sensitive Ambipolar Response towards Oxidizing NO<sub>2</sub> and Reducing NH<sub>3</sub> Based on Bis(phthalocyaninato) Europium Semiconductors

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High sensitive chemical sensors towards NO<sub>2</sub> and NH<sub>3</sub> based on the self-assembled nanostructures of the heteroleptic and homoleptic bis(phthalocyaninato) europium complexes with octanaphthoxy phthalocyaninato ligands named Eu(Pc)[Pc(ONh)<sub>8</sub>] (1) and Eu[Pc(ONh)<sub>8</sub>]<sub>2</sub> (2) [Pc = unsubstituted phthalocyaninate; Pc(ONh)<sub>8</sub> = 2,3,9,10,16,17,23,24-octanaphthoxy phthalocyaninate] have been developed. The good conductivity, high crystallinity and large specific surface area for the self-assemblies of 1 render it excellent sensing property for either electron-accepting gas NO<sub>2</sub> in 50–250 ppb range or electron-donating gas NH<sub>3</sub> in 2.5–12.5 ppm range due to the optimized molecular packing in the uniform-sized nanoparticles depending on the effective intermolecular interaction between double-decker molecules, among the best results of phthalocyanine-based chemical sensors for detection of NO<sub>2</sub> and NH<sub>3</sub> at room temperature. Interestingly, self-assemblies of 1 exhibited *n*-type response to NO<sub>2</sub> and *p*-type response to NH<sub>3</sub>, which is the first example of ambipolar charge-transporting gas sensors fabricated from singlecomponent organic semiconductors. However, the self-assemblies of 2 with sixteen bulky naphthoxy groups at the periphery of two Pc rings only present an *n*-type response to strong oxidant gas NO<sub>2</sub> in a relatively high concentration of 0.5–1.5 ppm, while are insensitive to weak reducing gas NH<sub>3</sub> due to the existence of great steric hindrance from bulky naphthoxy groups and more traps and/or defects in self-assemblies.

Keywords phthalocyanine, double-decker, gas sensor, self-assemblies, ambipolar semiconductor

## Introduction

Among functional molecular materials with large conjugated electronic molecular structures, phthalocyanines (Pcs) have received a particular attention due to their high thermal and chemical stabilities, great processability and rich substitution chemistry, leading to a large flexibility in tuning molecular structures and macroscopic properties.<sup>[1]</sup> In addition, the electrical conductivity of phthalocyanines, which are mostly *p*-type organic semiconductors, is known to vary substantially in the presence of oxidizing or reducing gases due to the change in charge carrier concentrations upon adsorption of gas molecules.<sup>[2,3]</sup> These materials have been widely explored as chemiresistive gas sensors since about twenty years ago.<sup>[4]</sup> For most of the applications, the properties of the molecular devices are closely related to the microstructures of the molecular materials. Many efforts have been made to clarify the relationship between the basic chemical structures, nanostructures self-assembled from Pc-based molecular materials and

the device performances. For example, Bouvet et al.<sup>[5]</sup> found that the vacuum evaporated film of *p*-type CoPc with well-organized structure was sensitive to NH<sub>3</sub> and  $O_3$ , while the drop-casting film of *n*-type sulfonated CoPc (s-CoPc) with low crystallinity was only sensitive to NH<sub>3</sub>. For two typical *n*-type organic semiconductors,  $F_{16}CuPc$  and  $(CN)_{16}CuPc$ , it has been revealed that the vacuum-deposited film of F<sub>16</sub>CuPc exhibited high organic field-effect transistor (OFET) electron mobility  $(0.03 \text{ cm}^2 \cdot \text{V}^{-1} \cdot \text{s}^{-1})$  and no OFET behavior was observed for the casting film of (CN)<sub>16</sub>CuPc having poor molecular ordering in the film.<sup>[6]</sup> However, it must be pointed out that optimizing the intermolecular interaction through the combination of molecular design with morphology-controlled self-assembly to improve sensing performance still remains a great challenge for chemists and material scientists.

Among the large family of the phthalocyanines, double-decker phthalocyanines have been subject to extensive investigations because of their radical nature, excellent molecular semiconducting property and possi-

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ble applications in gas sensing properties.[1,7-10] Very recently, solution-processed thin solid films of bis-(phthalocyaninato) holmium double-deckers bearing electron-withdrawing phenoxy substituents at the phthalocyanine periphery  $Ho(Pc)[Pc(OPh)_8]$ and  $Ho[Pc(OPh)_8]_2$  have been revealed to display the *p*-type and *n*-type current response to electron-donating gas NH<sub>3</sub> in 15-800 ppm range, respectively.<sup>[11]</sup> In light of these previous reports, we herein design and prepare the bulky naphthoxy substituted bis(phthalocyaninato) europium semiconductors  $Eu(Pc)[Pc(ONh)_8]$  (1) and  $Eu[Pc(ONh)_8]_2$  (2) [Pc=unsubstituted phthalocyaninate; $Pc(ONh)_8 = 2,3,9,10,16,17,23,24$ -octanaphthoxy phthalocvaninate], Scheme 1. The gas-sensing properties of the self-assembled nanostructures of 1 and 2 fabricated by a solution-processed quasi-Langmuir-Shäfer (QLS) method have been comparatively studied, which confirmed the strong effect of bulky naphthoxy substituents on intermolecular interactions and showed that self-assembled nanostructures of 1 have ambipolar response to both electron-accepting gas  $NO_2$  in 50-250 ppb range and electron-donating gas  $NH_3$  in 2.5-12.5 ppm range at room temperature.

Scheme 1 Schematic molecular structures of bis(phthalocya-ninato) europium double-decker complexes  $EuPc[Pc(ONh)_8]$  (1) and  $Eu[Pc(ONh)_8]_2$  (2)



## Experimental

## Chemicals

1,8-Diazabicyclo[5.4.0]undec-7-ene (DBU) was purchased from Aldrich. The solvent *n*-pentanol was distilled from sodium. DMF for voltammetric study was freshly distilled from CaH<sub>2</sub> under an atmosphere of nitrogen. The compounds of Eu(acac)<sub>3</sub>•H<sub>2</sub>O, EuPc(acac), 4,5-bis(naphthoxy)-1,2-dicyanobenzene were prepared according to the literature methods.<sup>[12-15]</sup>

### **Compound characterizations**

MALDI–TOF mass spectrum was taken on a Bruker BIFLEX III ultra-high-resolution Fourier transform ion cyclotron resonance (FT–ICR) mass spectrometer with alpha-cyano-4-hydroxycinnamic acid as matrix. <sup>1</sup>H NMR spectrum was recorded on a Bruker DPX 400 spectrometer in  $CDCl_3/[D_6]DMSO$  (1 : 1) in the presence of *ca.* 1% (by volume) hydrazine hydrate. Spectrum was referenced internally with residual DMSO ( $\delta$  2.49). The electrochemical cell comprised inlets for a glassy carbon disk working electrode of 3.0 mm in diameter and a silver-wire counter electrode. The reference electrode was Ag/Ag<sup>+</sup> (a solution of 0.01 mol/L AgNO<sub>3</sub> in acetonitrile), which was connected to the solution by a Luggin capillary, whose tip was placed close to the working electrode. Typically, a 0.1 mol·L<sup>-1</sup> solution of [Bu<sub>4</sub>N][ClO<sub>4</sub>] as the electrolyte in dried DMF containing 0.5 mmol·L<sup>-1</sup> of sample was purged with nitrogen for 5 min. After measurement, the result was corrected for junction potentials by being referenced internally to the ferrocenium/ferrocene (Fc<sup>+</sup>/Fc) couple [ $E_{1/2}$ (Fc<sup>+</sup>/Fc)=0.50 V vs. SCE].

### **Film measurements**

For characterizations, self-assemblies were fabricated by QLS method following previously published reference<sup>[16]</sup> onto the quartz substrate for UV-vis absorption spectra measurement, and onto the bare SiO<sub>2</sub>/Si substrate for X-ray diffraction (XRD) and atomic force microscope (AFM) measurements. UV-vis absorption spectra were recorded on a Hitachi U-4100 spectrophotometer. X-ray diffraction experiments were carried out on a Rigaku D/max-B X-ray diffractometer with copper (K $\alpha$ ) radiation ( $\lambda$ =1.5406 Å). Atomic force microscope images were collected under ambient conditions using the tapping mode with a NanoscopeIII/Bioscope scanning probe microscope from digital instruments.

#### **Electrical measurements**

The device of gas sensing was fabricated onto the Indium Tin Oxide (ITO) substrate which is composed of 10 pairs of interdigitated electrode array with the following dimensions: 125 mm electrode width, 75  $\mu$ m spacing, 6000  $\mu$ m overlapping length, and 20 nm electrode thickness. The fundamental electrical and sensor measurements were performed using a Keysight B2910A precision source/measure unit with an incorporated direct current voltage supply. The electrometer was controlled by Quick IV Measurement Software. Current-voltage (*I-V*) curves were registered in the voltage range of -10-+10 V. And for gas sensing experiments, the samples were constantly polarized at +5 V. All experiments have been conducted at least twice to ensure reproducibility.

Conductivity,  $\sigma$ , obtained by current-voltage (*I-V*) curves can be calculated by the following equation,<sup>[17]</sup>

$$\sigma = \frac{dI}{(2n-1)LhV} \tag{1}$$

where d is the interelectrode spacing, I the current, n the number of electrode digits, L the overlapping length of the electrodes, and h the electrode thickness as the film thickness exceeds that of the ITO electrodes in present case.

#### Flow system for chemical sensing

Sensing experiments were carried out in a cuboid Teflon sensor chamber (*ca.* 30 cm<sup>3</sup> as internal volume). The desired NO<sub>2</sub> or NH<sub>3</sub> concentrations were produced by diluting a mixture NO<sub>2</sub>/N<sub>2</sub> (50 ppm NO<sub>2</sub>, from Qingdao Ludong Gas., Ltd, China) or NH<sub>3</sub>/N<sub>2</sub> (500 ppm NH<sub>3</sub>, from Qingdao Ludong Gas., Ltd, China) or NH<sub>3</sub>/N<sub>2</sub> (500 ppm NH<sub>3</sub>, from Qingdao Ludong Gas., Ltd, China) with dry N<sub>2</sub> using two CS200 Mass Flow Controllers (total mass flow: 0.1 L•min<sup>-1</sup> for NO<sub>2</sub> or NH<sub>3</sub> and 2 L•min<sup>-1</sup> for diluent gas N<sub>2</sub>), respectively. All tests consisted of the exposure of the sensor to "a static atmosphere" followed by exposure to a known concentration of NO<sub>2</sub> or NH<sub>3</sub>.

## **Results and Discussion**

#### Synthesis and characterizations

The heteroleptic bis(phthalocyninato) rare earth complexes are usually prepared by the treatment of a rare earth acetate with two different lithium phthalocvaninates. To facilitate the synthetic yield and purification,  $EuPc[Pc(ONh)_8]$  (1) was obtained by cyclic tetramerization of the 4,5-bis(naphthoxy)-1,2-dicyanobenzene with the half-sandwich compound Eu(Pc)(acac) as the template in refluxing *n*-pentanol in the presence of 1,8-diazabicyclo[5.4.0]undec-7-ene (DBU) according to the literature methods.<sup>[13]</sup> A mixture of EuPc(acac) (91.4 mg), and 4,5-bis(naphthoxy)-1,2-dicyanobenzene (238 mg) in *n*-pentanol (3 mL) was refluxed for 3 h under a slow stream of nitrogen. After being cooled to room temperature, 20 mL *n*-hexane was added to the reactor. After standing for 6 h, the precipitate was filtered off, and then washed with *n*-hexane. The residue left was chromatographed on a silica gel column. Due to the poor solubility, a few drops of CH<sub>3</sub>OH must be added to the pure CHCl<sub>3</sub> [V(CHCl<sub>3</sub>)/V(CH<sub>3</sub>OH) = 99 : 1] as eluent. Repeated chromatography followed by recrystallization from CHCl<sub>3</sub> and *n*-hexane gave pure compound of green powder. Yield: 68 mg (30%). <sup>1</sup>H NMR (400 MHz, CDCl<sub>3</sub>)  $\delta$ : 11.32 (s, 8H, Pc-H<sub>a</sub>), 10.81 (s, 8H,  $Pc^*-H_{\alpha}$ ), 8.76 – 8.74 (s, 8H,  $Pc^*-H_{\beta}$ ), 8.76 (s, 8H,  $Pc^*-Nh-H_{B1}$ ), 8.63 (d, J=11.2 Hz, 8H, Pc-Nh-H<sub>B2</sub>), 8.55 (t, J=11.2 Hz, 16H, Pc-Nh-H<sub> $\beta$ 3, $\beta$ 4</sub>), 8.47 (d, J=16.0 Hz, 8H, Pc-Nh-H<sub>B7</sub>), 7.74 (d, J=2.0 Hz, 16H, Pc-Nh-H<sub>B5,B6</sub>). This newly prepared double-decker compound was characterized by MALDI-TOF spectrum with an isotopic cluster peaking at m/z 2314, Figure S1, similar with the Calcd result for  $C_{144}H_{80}EuN_{16}O_8$ ,  $[MH]^+$ , 2314.

The homoleptic bis(phthalocyaninato) rare earth(III) complex  $Eu[Pc(ONh)_8]_2$  (2) was synthesized by the self-cyclic tetramerization of phthalonitriles in the presence of metal salts according to the previously published procedures.<sup>[13,14]</sup> All other reagents and solvents were of reagent grade and used as received.

#### **Electrochemical properties**

The electrochemical behaviors of both doubledecker complexes were investigated by differential pulse voltammetry (DPV) in DMF. The double-decker complexes of 1-2 displayed one or two quasi reversible one-electron oxidation(s) and three quasi reversible one-electron reductions within the electrochemical window of DMF, Figure 1 and Table S1 (Support Information). The first oxidation and the first reduction processes involve the removal of an electron from and the addition of an electron to the singly occupied molecular orbital, respectively. It is found that the halfwave potentials of the first oxidation as well as the first and second reductions of two compounds correlate with the number of the peripheral octanaphthoxy substituents, which are shifted in the anodic direction along with the increase of the number of octanaphthoxy substituents from 1 to 2, indicating that incorporation of the sixteen naphthoxy groups on two phthalocyanine rings makes  $Eu[Pc(ONh)_8]_2$  (2) slightly harder to oxidize and easier to reduce than the analogous compound Eu(Pc)- $[Pc(ONh)_{8}]$  (1) with eight naphthoxy groups on the phthalocyanine ring. These results are consistent with those findings for the phenoxy and thiophenoxy-substituted bis(phthalocyaninato) metal complexes<sup>[13]</sup> and octyloxycarbonyl-, F- and CN-substituted phthalocyaninato metal complexes,<sup>[18]</sup> which clearly reveal the electron-withdrawing nature of these naphthoxy substituents on the phthalocyaninato ligand. In the present



Figure 1 Differential pulse voltammetry (DPV) of 1 (A) and 2 (B) in DMF containing 0.1 mol·L<sup>-1</sup> [Bu<sub>4</sub>N][ClO<sub>4</sub>] at a scan rate of 20 mV/s vs. SCE.

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case, the HOMO and LUMO energy levels of 1-2 were estimated to be about -4.85 and -3.61 eV for 1 and -4.95 and -3.73 eV for 2, on the basis of their half-wave potentials of the first and second reductions (vs. SCE).<sup>[11,19,20]</sup> The HOMO and LUMO energies of 1 -2 just simultaneously meet the energy range required for *p*-type and *n*-type organic semiconductors,<sup>[21-23]</sup> suggesting the ambipolar organic semiconducting nature of these double-decker complexes.

#### Film morphologies and microstructures

The self-assemblies of  $EuPc[Pc(ONh)_8]$  (1) and  $Eu[Pc(ONh)_8]_2$  (2) were prepared by a solution-based OLS method.<sup>[16,25]</sup> The quality of the self-assemblies</sup>could be assessed using UV-vis absorption spectra, X-ray diffraction pattern (XRD), and atomic force microscopy (AFM) techniques. As shown in Figure 2, typical UV-vis absorption spectra of heteroleptic and homoleptic bis(phthalocyaninato) rare earth(III) complexes were observed, with the peaks in the 470-490 nm region associated with a free radical structure of rare earth bisphthalocyanines.<sup>[11,26]</sup> The spectra also show the intense Soret bands in the range of 320-360nm, and a O-band, the most intense, at 675 nm for 1, and 680 nm for 2, which has been attributed to a  $\pi$ - $\pi$ \* transition, respectively. With the comparision of heteroleptic compound 1 in solution, slightly red-shifts take place in both Q and Soret bands for the naphthoxy-



**Figure 2** UV-vis absorption spectra of compounds **1** (A) and **2** (B) in DMF solution (solid line) and self-assemblies obtained by QLS method (dot line).

substituted homoleptic double-decker compound 2. This result is in line with the behavior found from heteroleptic and homoleptic bis(phthalocyaninato) holmium(III) counterparts attaching phenoxy substituents with a slight electron-withdrawing nature.<sup>[11,13]</sup> Obviously, these optical transitions have a significant contribution from the frontier molecular orbitals delocalized mainly on the two Pc rings, which makes them more susceptible to functionalization by attaching electron-withdrawing groups to the conjugated Pc cores.<sup>[11,27]</sup> However, after being fabricated into the self-assemblies. the main Q bands were broadened and red-shifted from 675 to 679 nm for 1 and from 680 to 686 nm for 2, similarly the Soret bands were also red-shifted to 329 nm and 331 nm, respectively. On the basis of Kasha's exciton theory,<sup>[28]</sup> red shift in the main absorption bands of self-assemblies is a typical sign of that the doubledecker molecules are enforced to adopt the J aggregation mode (where molecules stack by a head-to-tail arrangement). These results are in contrast to those found for phenoxy-substituted heteroleptic and homoleptic bis(phthalocyaninato) holmium(III) counterparts with *H*-aggregation mode (where molecules stack by a face-to-face arrangement), suggesting that bulky naphthoxy substituted side groups would favor packing of adjacent molecules in parallel so that the bulky side groups avoid proximity and minimize steric hindrance. It should be noted that along with the increase in the number of bulky naphthoxy substituents from compound 1 to 2, the extent of red-shift for the main Q band only slighly increases, implying the steric hindrance of bulky naphthoxy substituents has a great influence on the intermolecular interaction in the self-assemblies.

To obtain information of the intermolecular interaction between adjacent molecules, XRD spectra of the self-assemblies of 1 and 2 were recorded in Figure 3. As revealed, in the low angle range, the XRD diagram of the self-assemblies formed from 1 shows a strong refraction peak at  $2\theta$ =4.32° (corresponding to 2.05 nm), which is ascribed to the diffraction from the (001) plane. In addition, two clear diffraction peaks at  $2\theta = 6.13^{\circ}$ (1.44 nm) and  $2\theta = 9.78^{\circ}$  (0.90 nm) are also presented, respectively, originating from the (010) and (100) planes. It is noteworthy that an additional diffraction appeared at 0.32 nm in the wide angle region, which should be attributed to the  $\pi$ - $\pi$  stacking distance between the phthalocyanine rings without the naphthoxy group of neighboring double-decker molecules along the direction perpendicular to the phthalocyanine rings.<sup>[29,30]</sup> These three diffraction peaks in the low angle range should be assigned to diffraction from a parallelepiped lattice with the cell parameters of a=2.05 nm (length), b=1.44 nm (width) and c=0.90 nm (thickness), the inset of Figure 3A, formed from edge-to-edge (J-type) stacking of two double-decker molecules depending on the  $\pi$ - $\pi$  stacking and the van der Waals intermolecular interactions between neighboring double-decker molecules on the basis of the energy-optimized, dimensional



Figure 3 X-ray diffraction patterns of the self-assemblies of compounds 1 (A) and 2 (B). The insets are schematic packing modes.

size for the double-decker EuPc[Pc(ONh)<sub>8</sub>] (1) [2.86 nm  $(diameter) \times 0.60$  nm (height)] using PCMODEL for windows Version 6.0, Serena Software. As for the self-assemblies of Eu[Pc(ONh)<sub>8</sub>]<sub>2</sub> (2), the XRD diagram shows only one peak at 1.99 nm in the low angle range, Figure 3B, corresponding to the thickness of one layer of the film, suggesting the lamellar film-structure formed.<sup>[31,32]</sup> As a result, a structural transformation from the parallelepiped lattice to the lamellar structure is achieved by increasing the number of naphthoxy groups attached onto the periphery of the Pc rings for the double-decker complex. Judging from the diagonal dimension of the compound 2 molecule obtained on the basis of the energy-optimized molecular structure, 2.86 nm, the orientation angle between the phthalocyanine ring in the double-decker molecule and substrate surface is calculated to amount to ca. 44.1°. This is in line with the calculated result (43.9°) based on the polarized UV-vis spectroscopy according to a literature method,<sup>[33]</sup> Figure S2 and Table S2 (Supporting Information). Consequently, the double-decker molecules of 2 are oriented nearly parallel to the substrate surface in J-aggregation mode with an "edge-on" conformation of the doubledecker molecules in the film (inset of Figure 3). This is in line with the UV-vis measurement results (vide su*pra*). Moreover, the intensity of the (001) peak at  $2\theta =$ 4.32° for 1 increased remarkably, compared to that for 2, indicating a higher crystallinity for self-assembled nanostructure of 1. The favorable  $\pi$ - $\pi$  stacking interaction among the double-decker molecules together with

enhanced crystallinity for the self-assemblies of **1** is believed to provide the  $\pi$  electrons (or holes) with an extensive area for delocalization, thus favoring charge transport. In the case of the self-assemblies of **2**, only (001) diffraction peak is observed without the presence of an obvious  $\pi$ - $\pi$  stacking feature, which will negatively impact on the charge transports, as discussed further below.

AFM images can provide more information on the aspect of the self-assemblies and therefore allow us to correlate the morphology and electrical properties. As revealed in Figure 4A, the images for the self-assemblies of 1 show uniform small grain crystallites with approximately 100 nm in diameter. A root-mean-square (Rrms) roughness value of 1.06 nm is revealed, implying a typical smooth surface for the self-assemblies of 1. On the other hand, some disconnected domains with ca.  $(150\pm15)$  nm in size were observed for the self-assemblies of 2, Figure 4B, giving a relatively high Rrms value of 2.65 nm. The more uniform grain size and much lower Rrms value for the self-assemblies of 1 compared to those of 2 is expected to cause fewer traps and/or defects localized around grain boundaries, thus improving the electrical conductivity and gas sensing as detailed below.



Figure 4 AFM images and corresponding 3D enlarged images for the self-assemblies of 1 (A) and 2 (B).

#### **Electrical and sensing properties**

The electron conductivity measurements were conducted by *I-V* curves for the self-assemblies of 1-2. As shown in Figure 5, both two devices exhibit similar Ohmic behavior at low bias voltage, confirming the good electrical contact between the organic self-assemblies and electrodes. However, current response decreased significantly along with the increase in the number of naphthoxy groups of the macrocycles from 1 to 2, with the calculated electronic conductivity about  $(2.07\pm0.5)\times10^{-5}$  and  $(6.79\pm0.5)\times10^{-6}$  S•cm<sup>-1</sup> for the

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aggregates of 1 and 2, respectively. Comparing with that of 2, the improved conductivity of self-assemblies of 1 might be attributed to both readily  $\pi$ -stacks with adjacent EuPc[Pc(ONh)<sub>8</sub>] molecules and higher ordered molecular arrangement from reducing the steric hindrance of bulky naphthoxy substituents in addition to the fewer traps and/or defects in the self-assemblies of 1. This result suggested that the number of substituent groups plays a critical role in tuning the intermolecular stacking, and then impacts on electrical conductivity of materials.



Figure 5 *I-V* curves measured on the self-assemblies of 1 and 2.

To assess the applicability of gas sensing of doubledecker complexes 1-2, the devices fabricated based on the self-assemblies of 1-2 were exposed to different concentration of electron-accepting gas NO<sub>2</sub> (50-250 ppb for 1, 0.5-1.5 ppm for 2) and electron-donating gas NH<sub>3</sub> (2.5-12.5 ppm for 1, 25-250 ppm for 2) in  $N_2$  atmosphere at room temperature. The sensing performances were studied with a duty cycle where the dynamic exposure period was fixed at 5 min for NO<sub>2</sub> and 3 min for NH<sub>3</sub>. The responses to NO<sub>2</sub> and NH<sub>3</sub> for the self-assemblies of 1 and 2 were recorded, Figure 6. As can be seen qualitatively in Figure 6A, the current of the self-assemblies of 1 decreased during exposure to 50 -250 ppb NO<sub>2</sub> and increased during recovery, which is consistent with the sensing behavior reported previously for *n*-type semiconductors.<sup>[34,35]</sup> Indeed, NO<sub>2</sub> is a strong oxidizing agent, which generates additional positive charge carriers in  $EuPc[Pc(ONh)_8]$ , and in turn leads to a decrease in electron concentration (majority charge carriers for *n*-type semiconductors). Interestingly, upon exposure to the electron-donating gas  $NH_3$  in 2.5-12.5 ppm range, the conductivity of the self-assemblies of 1 still decreased, Figure 6B, as the behaviors for *p*-type semiconductors.<sup>[5,11]</sup> When electron-donating  $NH_3$  molecules are absorbed on the surface of the *p*-type semiconductors, the hole concentration of the surface will be weakened since NH<sub>3</sub> traps majority charge carriers (holes), resulting in the decrease in conductivity.<sup>[36]</sup> Therefore, ambipolar (*n*- and *p*-type) response towards oxidizing NO<sub>2</sub> and reducing NH<sub>3</sub> was revealed for the self-assemblies of 1. To the best of our knowledge, this

is the first example of ambipolar charge-transporting gas sensing devices fabricated from single-component organic semiconductors. However, when the self-assemblies of **2** were exposed to 0.5-1.5 ppm NO<sub>2</sub>, the conductivity significantly decreased, exactly as expected for an *n*-type semiconductor, Figure 6C. In addition, the sensors fabricated from the self-assemblies of **2** were totally insensitive to both 50–250 ppb NO<sub>2</sub> applied for **1** and 25–250 ppm NH<sub>3</sub> (whose concentration is above ten times higher than that used for **1**), Figure 6D. The poor sensing response of **2** should mainly be ascribed to the existence of steric hindrance, thus either a small amount of NO<sub>2</sub> or much more NH<sub>3</sub> molecules adsorbed on the surface of the self-assemblies have little contributions to the conductivity.

In order to quantitatively analyze the sensor responses, the percent current change was calculated for each concentration, as follows:

% current change = 
$$[(I_f - I_0)/I_0] \times 100$$
 (2)

where  $I_0$  is the current at the start of an exposure/rest cycle and  $I_{\rm f}$  is the current at the end of the exposure period. The values are designated as the sensor response at given NO<sub>2</sub> and NH<sub>3</sub> concentrations as depicted in Figures 6E and 6F. It can be seen that for both self-assemblies of 1 and 2, the responses are fitting nonlinear curves with respect to analyte concentration (in general  $R^2 \ge 0.98$ ). From the equations of the curves as shown in Figures 6E-F, the slopes are decreased gradually as the concentrations of the analytes increase, which mean both compounds 1 and 2 are more sensitive to low concentrations of both NO2 and NH3, which is important for the application as gas sensing and/or detector. However, further increasing the concentration of the analyte led to a lower sensitivity. Specially, the response gradually approached the saturated status to the relatively high concentration of analyte. This implied that the much more analyte molecules have completely filled the adsorption sites of the semiconducting layer and the redundant gases have little contribution to the conductivity of the films. Similar behaviors have been reported for MPc-based sensors.<sup>[11,36,37]</sup> By comparing the value of the slope, EuPc[Pc(ONh)8] is more sensitive to analytes than  $Eu[Pc(ONh)_8]_2$ . The results show that both conductivity and microstructure of self-assemblies have significant effect on the sensing characteristics. The better electric conductivity, higher crystallinity and larger specific surface area for the self-assemblies of 1 than those of 2 render it excellent sensing property for either electron-accepting gas NO<sub>2</sub> in the 50-250 ppb range or electron-donating gas  $NH_3$  in the 2.5-12.5 ppm range due to the optimized molecular packing in the uniform-sized nanoparticles depending on the effective intermolecular interaction between double-decker molecules. The results are inter esting since we can easily detect concentration of NH<sub>3</sub> well below the requirements for detection in industrial wastes control (~10



**Figure 6** The time-dependent current plots (A-D) and relative response of sensors as a function of the analyte concentration (E-F) for the self-assemblies of 1 and 2 exposed to NO<sub>2</sub> and NH<sub>3</sub> (A, B for 1, C, D for 2) at varied concentration in N<sub>2</sub> atmosphere.

ppm NH<sub>3</sub>) at room temperature. Furthermore, the lowest NO<sub>2</sub> detection limit of 50 ppb has also met the required standard for national primary ambient air quality ( $\sim$ 50 ppb NO<sub>2</sub>) in China.

## Conclusions

Heteroleptic and homoleptic bis(phthalocyaninato) europium semiconductors  $Eu(Pc)[Pc(ONh)_8]$  (1) and  $Eu[Pc(ONh)_8]_2$  (2) with different number of bulky electro-withdrawing naphthoxy substituents at the periphery of phthalocyanine rings are designed and synthesized. Their LUMO and HOMO levels are successfully tuned

into the required range for ambipolar organic semiconductors. Using solution-processed QLS method, selfassemblies with different microstructure and morphology were obtained, with an optimized molecular packing in the uniform-sized nanoparticles and high crystallinity for 1 and disconnected nanograin domains with large gaps and cracks for 2, depending on the competition and cooperation between the intermolecular  $\pi$ - $\pi$ interaction, the van der Waals forces and steric hindrance of the bulky naphthoxy groups between neighboring double-decker molecules. The self-assemblies of 1 are more sensitive to both NO<sub>2</sub> and NH<sub>3</sub> relative to

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those of **2**, with the detection limit as low as 50 ppb for NO<sub>2</sub> and 2.5 ppm for NH<sub>3</sub>, respectively, implying inherently good gas sensing properties of the bis(phthalocyaninato) rare earth double-decker complex. Meanwhile, self-assemblies of **1** exhibited *n*-type response to NO<sub>2</sub> and *p*-type response to NH<sub>3</sub>, while only *n*-type performance was obtained to strong oxidant gas NO<sub>2</sub> for the self-assemblies of **2**. The present work opens a new way for the preparation of high-performance ambipolar chemical sensors through the combination of molecular design and solution-based self-assembled technique.

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